Article

Study of Charge Compensation During the Redox Process of Self-Doped Polyaniline in Aqueous Media

Hamilton Varela^a, Roberto M. Torresi^{a*} and Daniel A. Buttry^b

^aInstituto de Química de São Carlos, Universidade de São Paulo, CP 780, 13560-970, São Carlos - SP, Brazil ^bDepartment of Chemistry, University of Wyoming, Laramie, Wyoming, 82071-3838, USA

Um dos problemas mais importantes associado à utilização da polianilina como material catódico em baterias recarregáveis de lítio está relacionado à degradação na densidade de energia devido à participação massiva de ânions durante o processo de compensação de cargas. Este trabalho descreve a síntese de uma polianilina auto-dopada, poli-(anilina-co-N-ácido propanossulfônico-anilina), e avalia suas propriedades em meio aquoso, considerando o aumento na participação de prótons no mecanismo de eletroneutralização. A caracterização foi feita utilizando análise elementar e espectroscopias no infravermelho e ultravioleta-visível. As propriedades eletroquímicas foram investigadas com a microbalança eletroquímica a cristal de quartzo e voltametria cíclica. Os resultados obtidos mostram que a participação de prótons no processo redox da poli-(anilina-co-N-ácido propanossulfônico-anilina) aumentou, em relação à polianilina.

One of the most important problems associated with use of polyaniline as a cathode material in rechargeable lithium batteries is related to energy density degradation due to the predomination of anion participation in the charge compensation process. This work describes the synthesis of a self-doped polyaniline, poly-(aniline-co-N-propanesulfonic acid-aniline), and evaluates its properties in aqueous acid solutions, with special attention to the increase of proton participation in the electroneutralization mechanism. The characterization was carried out using elemental analysis, FTIR and UV-vis spectroscopies. Electrochemical properties were investigated with the electrochemical quartz crystal microbalance and cyclic voltammetry. The results obtained show that proton participation to polyaniline.

Keywords: poly (aniline-co-N-propanesulfonic acid-aniline), polyaniline, electrochemical crystal quartz microbalance, charge compensation process, ECQM.

Introduction

Polymers with π -conjugated bonds in the chain show uncommon properties like low energy electronic transitions, low ionization potential and high electron affinity. These properties result in polymers that can be oxidized and reduced easily. In order to compensate the charge produced in the polymer matrix by such redox processes, there is usually incorporation/expulsion of ionic species (dopants). Among conducting polymers, polyaniline (PANI) has a prominent role due to its potential applications in energy storage, electrochromic devices, chemical sensors and a series of other electrochemical devices^{1,2}. PANI has also been evaluated for use as cathode for rechargeable lithium batteries³.

The participation of ionic species in the insertion/expulsion process and their influence on the charge compensation mechanism needs to be investigated because this knowledge is of critical importance in the evaluation of PANI as a cathode material for secondary batteries. Ionic transport in the film, necessary to maintain its electroneutrality, has been thoroughly investigated in the case of aqueous solution⁴⁻⁷. The main problem related with PANI as a cathode for a lithium battery is the majority participation of anions in the charge compensation process⁸. This fact produces a decrease in the energy density, leading to a power capacity lower than 30 Wh kg⁻¹ for the PANI-Li system⁹. It is obvious that an increase in the energy density can be expected if the polymer uses only cations for the charge compensation process¹⁰. This conclusion is related to different types of configuration for batteries using conducting polymers as

e-mail: Torresi@iqsc.sc.usp.br

cathode. Among the six different configurations for batteries with conducting polymer cathodes11, three of them are based on the principle that the cathode uses the cation produced in the anode oxidation to compensate charge. This kind of configuration has the advantage that it does not use a great amount of excess supporting electrolyte so, thin electrolyte films can be used. As a consequence, a diminution in the mass and volume of the battery is obtained. This produces an increase in the specific properties of the battery. Figure 1 shows two different configurations. In Figure 1a, the polymer used as the cathode material might be a material such as PANI, for example, where anions play an important role in charge compensation process and the battery will require an electrolytic reservoir to store anions. In Figure 1b, a polymeric cathode is used where cations are responsible for neutralization process. In this case, the cathode can use the cations produced in the anode, and the described reservoir can be replaced by a thin membrane for cation transport.



a



Figure 1. Different configurations for secondary batteries with eletronic conducting polymer as cathode. (a) Charge compensation process carried out by anions and (b) Charge compensation process carried out by cations. Adapted from reference 11.

It is very important to establish the nature of the ionic exchange and solvent participation in the redox process of conducting polymers in order to choose the best design of the battery, improving in this way their properties. Two alternatives to attempt an increase of the specific properties of PANI for lithium batteries have been studied. First, derivatives obtained by the modification of the original monomer; second, the composites like PANI/NAFION12 or PANI/DMcT¹³. In this work, the properties of self doped, poly - (aniline - co - N - propanesulfonic acid - aniline), PAPSAH, were investigated in aqueous media by the use of electrochemical crystal quartz microbalance (ECQM) in conjunction with cyclic voltammetry (CV) experiments. In addition, the synthesis and characterization of PAPSAH are described. For comparison, electrochemically formed PANI films were also investigated. The main purpose is to obtain information about the influence of anionic group (propanesulfonic acid) linked to the polymer chain on the charge compensation process neutralizing the positive charge on the PANI chains.

Experimental

PAPSAH films were formed by solvent evaporation and PANI films were electrochemically deposited onto Au electrodes supported onto 6 MHz AT cut quartz crystals that were 26 mm in diameter. A platinum wire was used as counter electrode and all potentials are referred to the saturated calomel electrode (SCE). Both sides of the quartz crystals were coated with gold by vacuum deposition but only one of the faces was exposed to the electrolyte solution (active area of 0.2 cm²). The resonance frequency shift was measured with a HP-5370B Universal Time Counter and electrochemical measurements were performed with a FAC 2001 potentiostat/galvanostat.

The electrochemical polymerization of aniline was carried out by cycling the potential between -0.2V and 0.75V (versus SCE) at 50 mV/s in a solution of 1mol dm⁻³ HCl + 0.5M C₆H₅NH₂. All resulting polyaniline films presented good adherence and homogeneity. Films were grown until a net mass gain of 28 mg cm⁻² was reached corresponding to a film thickness of 55 nm, determined using a profilometer. After deposition, films were washed carefully with purified water. Electrodes were placed in a usual three-compartment electrochemical cell containing monomer free 1 mol dm⁻³ HCl or HClO₄ aqueous solutions.

The method employed for the chemical synthesis of PANI was previously described by Huang et al¹⁴. An aqueous solution of $(NH_4)_2S_2O_8$ was added slowly to a solution of aniline dissolved in 1.0 mol dm⁻³ aqueous HCl at 5°C. After 1 h the precipitate which had formed was re-

moved by filtration, washed repeatedly with 1.0 mol dm-3 HCl and dried under dynamic vacuum for 48 h. The material thus obtained was identified as emeraldine hydrochloride. The emeraldine hydrochloride was converted into emeraldine base by stirring with 0.1 mol dm-3 solution of NH₄OH for several h. The material was dried under dynamic vacuum for 48 h. The lower molar mass species were extracted with CH3CN until the extract was colourless. The PANI (2.4 g) obtained, in the blue emeraldine base form reacted with an excess NaH (1.1 g) in dimethyl sulfoxide (DMSO), at 45°C under nitrogen atmosphere for 6 h. The result was a black green solution, in which the original emeraldine base should have been converted to the deprotonated form of the emeraldine base. The resulting solution was reacted with an excess of 1,3 - propanesulfone (4.25 g) for 20 h. at room temperature, yielding a black-blue solution. This solution was precipitated with 1 mol dm-3 HCl aqueous solution, the green solid obtained (HCl doped PAPSAH) after filtration was washed with a large amount of acetonitrile and then undoped with NaOH aqueous solution to give a blue solution. This solution, after filtration, was purified by dialysis with semipermeable membrane (Sigma D - 9652, molar mass cutoff of 12,400 g mol-1) in purified water (Milli - Q, Millipore), which was replaced with fresh water frequently during 2 days. The poly(aniline-co-sodium N-propanesulfonate-aniline, PAPSANa) aqueous solution obtained was concentrated in a vacuum evaporator (MA 120, Marconi) at a temperature below 50°C. The polymer PAPSANa was then converted to poly(aniline-co-N-propanesulfonic acid-aniline, PAPSAH) shown in Figure 2, by exchanging Na+ for H+ in a H+-type ionexchangeresin (Dowex 50WX4-100, Sigma) to give a PAPSAH green coloured aqueous solution (doped state). The modified electrode was obtained by casting this green solution onto the working electrode. The deposited mass was typically 30 µg cm-2.



Figure 2. PAPSAH structure.

Results and Discussion

PAPSAH characterization

The elemental analysis results are shown in Table 1. The S/N ratio obtained for PAPSAH was 0.55, close to 0.48 observed in previous results¹⁵, showing that about 50 % of total nitrogen in the chains are linked to alkane sulfonic acid.

Table	1.	Elemental	analysis	results
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Element	Ν	С	Н	S
Mass Percentual	5.86	44.78	4.23	7.41

Figure 3 shows the FTIR spectra for PANI and PAPSAH. The bands obtained are compared to literature data¹⁶⁻¹⁹ in Table 2 showing that the present results agree with those obtained by others authors. The PAPSAH IR spectrum (Figure 3b) shows that the intensity ratio of the peaks at ~1585 cm⁻¹ and ~1500 cm⁻¹ is a qualitative measurement of the oxidation state of the polymer and, in both cases, these ratios show the same oxidation state, indicating an oxidation level of ca 0.8. The presence of the absorption bands at 1150 and 1034 cm-1 for PAPSAH is characteristic of the asymmetric and symmetric O=S=O stretching vibrations, respectively. Both polymers (Figures 3a and 3b) present absorption bands at 826 and 821 cm-1, characteristic of the C-H out-of-plane bending vibrations of 1,4 - disubstituted benzene rings. This shows that the reactions used to produce PAPSAH do not lead to ring substitution at other positions. The characteristic band of the secondary aryl amine bending mode (1310 cm-1) confirms that the alkanesulfonic acid groups are linked to the amine nitrogens.

Figure 4 shows UV-vis spectra for PAPSAH aqueous solutions in base and acid forms. The insulator/conductor transition can be discussed in terms of polaron bands, related to base/acid form. The bands located at 400 and 860 nm (3,10 and 1.44 eV) are attributed to polaron/bipolaron appearance, and the disappearance of 570 nm (2.17 eV) band is related to quinoid rings¹⁰. These results suggest that when PAPSAH is converted from the base to the acid form, there is a strong electronic delocalization, leading to an increase in the mobility of charge carriers.

Redox process (Figure 5a) shows the j/E and $\Delta m/E$ potentiodynamic profiles of PAPSAH in HCl electrolytic solution at 50 mV s⁻¹. It can be seen that at the beginning of the oxidation process (between 0 and 0.2 V) there is a mass diminution, and at more positive potentials than 0.2 V, the mass increases. This behaviour clearly shows that between 0 and 0.2 V, the electroneutralization in the oxidation

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Table	2.	Characteristic	bands	observed	in	the	FTIR	spectrum
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Characteristic band	Pani / cm ⁻¹		PAPSAH / cm ⁻¹		
	Exptl.	Literature ^{15,16}	Exptl.	Literature ^{15,16}	
C=C quinoid streching	1583	1582	1583	1592	
C=C benzenoid streching	1491	1485	1498	1501	
C-N secondary aryl amine bending	1300	1311	1310	1312	
O=S=O asymmetric streching			1150	1167	
O=S=O symmetric streching			1034	1031	
C-H out-of-plane bending	826	803	821	823	

process is mainly carried out by proton expulsion, while the incorporation of chloride anions becomes more important at more positive potentials. During the reduction process a mass diminution is observed as a consequence of chloride anion expulsion and proton incorporation.



Wavenumber / cm⁻¹

Figure 3. FTIR spectra. (a) PANI and (b) PAPSAH.



Figure 4. UV-vis spectra for PAPSAH in acid (_____) and base (.....) forms.



Figure 5. (a) j/E (_____) and $\Delta m/E$ (_____) potentiodynamic profiles and (b) proton+solvent (_____) and anion fluxes (_____) for PAPSAH in HCl 1mol dm⁻³ electrolytic solutions. v = 50 mV s⁻¹.

In order to analyze the EQCM results in a quantitative form, it is possible to write the global mass change caused by participation of protons, anions and solvent molecules as:

$$\Delta m_{(E)} = W_{H^+} \xi_{H^+(E)} + W_{A^-} \xi_{A^-(E)} + W_S \xi_{S(E)}$$
(1)

where W is the molar mass of protons (H⁺), anions (A⁻) or solvent molecules (s) and ξ is the number of moles of of protons, anions or solvent molecules. Considering electroneutralization conditions, the charge involved in the redox process as a function of the number of moles of ionic species can be written as

$$q_{(E)} = -F\xi_{H^+(E)} + F\xi_{A^-(E)}$$
(2)

where F is the Faraday constant. The charge for protons

and anions was considered 1 and -1, respectively. Combining equations 1 and 2, the contribution of anions, xA can be calculated together with solvent participation as a function of potential

$$\xi_{A^{-}(E)} + \frac{W_{S}}{W_{HA}} \xi_{S(E)} = \frac{\Delta m_{(E)}}{W_{HA}} + \frac{W_{H^{+}}}{W_{HA}} \frac{q_{(E)}}{F}$$
(3)

In the same way, can be obtained for the protons,

$$\xi_{H^{+}(E)} + \frac{W_{S}}{W_{HA}}\xi_{S(E)} = \frac{\Delta m_{(E)}}{W_{HA}} - \frac{W_{A^{-}}}{W_{HA}}\frac{q_{(E)}}{F}$$
(4)

where HA corresponds to each acid used.

By differentiation of equations 3 and 4 one obtains the flux of protons and anions with a contribution of solvent as a function of the current density and mass flux

$$\frac{d\left(\xi_{H^{+}(E)} + \frac{W_{S}}{W_{HA}}\xi_{S(E)}\right)}{dt} = \frac{1}{W_{HA}}\frac{d\left(\Delta m_{(E)}\right)}{dt} - \frac{W_{A^{-}}}{W_{HA}}\frac{j_{(E)}}{F}$$
(5)

and

$$\frac{d\left(\xi_{A^{-}(E)} + \frac{W_{S}}{W_{HA}}\xi_{S(E)}\right)}{dt} = \frac{1}{W_{HA}}\frac{d\left(\Delta m_{(E)}\right)}{dt} + \frac{W_{H^{+}}}{W_{HA}}\frac{j_{(E)}}{F}$$
(6)

Negative flux values refer to ejection and positive values to incorporation of species in the film. Solvent participation in the process is assumed to be contributing to the observed mass responses when the sign of the flux is contrary to expectations

Figure 5b shows the proton and chloride fluxes with the solvent contribution as a function of potential. During the oxidation process, the proton flux shows a maximum at ca 0.18 V while the chloride flux shows a maximum at ca 0.3 V. This fact clearly demonstrates not only the majority participation of protons at the beginning of the oxidation reaction, but also that the compensation by anions is more important at higher potentials.

Figure 6a shows the j/E and $\Delta m/E$ potentiodynamic profiles of PAPSAH in HClO4 electrolytic solution at 50 mV s⁻¹. The phenomenological behaviour observed is similar to that obtained for HCl solutions. However, at the beginning of the oxidation process the loss of mass is smaller than for HCl solution (see Figure 5a). This fact is a consequence of the greater molar mass of perchlorate, that is to say, even with a reasonable amount of proton expulsion occuring simultaneously with a small amount of perchlorate incorporation, the mass diminution is smaller than in the case of chloride anions. Figure 6b shows the proton and perchlorate fluxes with the solvent contribution as a function of potential. The behavior is similar to that observed for chloride solution.



Figure 6. (a) j/E (_____) and $\Delta m/E$ (_____) potentiodynamic profiles and (b) proton+solvent (_____) and anion fluxes (_____) for PAPSAH in HClO₄ 1 mol dm⁻³ electrolytic solutions. v = 50 mVs⁻¹.

The proton and anion fluxes also have a solvent contribution. In order to analyze this influence, the ratio of total mass/charge for chloride and perchlorate was calculated. $(\Delta m_t/q)$ for chloride solutions was 0.22 g/C and for perchlorate solutions was 0.60 g C⁻¹. The ratio between both values chloride/perchlorate is 0.37 and the ratio between the molar mass of both anions is 0.36. This shows that the mass changes can be completely accounted for using the molar masses of the anions, and suggests that there is not an important exchange of water molecules during the redox process.

In order to compare the results obtained with PAPSAH, an experiment with PANI was carried out in HCl solution. Figure 7a shows the j/E and Δ m/E potentiodynamic profiles at 50 mV s⁻¹. The mass profile is similar to that obtained for PAPSAH, showing a mass increase during the oxidation process and a mass diminution during the reduction process. Figure 7b shows the fluxes of ionic+solvent species as a function of potential. This Figure shows that



Figure 7. (a) j/E (_____) and Dm/E (.....) potentiodynamic profiles and (b) proton+solvent (_____) and anion fluxes (.....) for PANI in HCl 1mol dm⁻³ electrolytic solutions. v = 50 mVs⁻¹.

In order to compare the contribution of anions in both polymers, the transport number of anions as a function of potential for the oxidation process is shown in Figure 8. The transport number can be calculated using equations 5 and 6 in the case that there is not an important contribution of solvent, so the transport number of protons (t_{H+}) is

$$t_{H^{+}} = \frac{\frac{d\xi_{H^{+}}}{dt} x F}{j}$$
(7)

and the transport number of anions is

$$t_{A^{-}} = \frac{\frac{d\xi_{A^{-}}}{dt} x F}{j}$$
(8)



Figure 8 Anion transport number of PANI (_____) and PAPSAH (_____) in HCl 1mol dm⁻³ electrolytic solutions. Data calculated from Figures 5 and 7.

The results show that for potentials less positive than 0.2 V the anion contribution to charge compensation is six times greater for PANI²⁰ than for PAPSAH. Anion contribution increases for both polymers as a function of potential, and finally anions are totally responsible for the charge compensation process (transport number equal to one) in both polymers at sufficiently positive potentials. In this way, funtionalization of the PANI backbone with anionic groups like sulfonate leads to an important improvement of proton contribution to the charge compensation process in specific potential ranges.

Conclusions

The characterization of PAPSAH is in agreement with reported data in the literature, considering both physical and electrochemical properties. The strategy of comparing the chemically modified polymer PAPSAH with electrochemically formed PANI was very useful as a guide in the properties evaluation. The cation participation during the redox process was increased by between two and six times, depending on the potential region when the water soluble polymer PAPSAH was used. For potentials more positive than 0.25 V, the contribution of anions is the same for both polymers and the transport number equal to 1, that is to say, that anions are totally responsible for charge compensation process at these potentials.

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